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A FIRST-PRINCIPLES STUDY EFFECT PRESSURE OF ELECTRONIC AND OPTICAL PROPERTIES OF R-TiO₂

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Abstract

The band structure, density of state, and optical properties of TiO₂ rutile were studied using first principles calculations within the framework of density functional theory using the generalized gradient approximation (GGA-RPBE) at both zero and high pressures. we used the Birch-Murnaghan equation of state calculate of volume and bulk modulus by approximation that we mentioned above and our results were compared with previous theoretical and experimental data, we noticed a good agreement between the results. The band gap of 2.098 eV (GGA) remains unchanged when the pressure is increased from 0 to 10 GPa, indicating an underestimation. The reduction in volume and lattice constants with increasing pressure is responsible for the decrease in band gap. There is a good agreement between the experimental results and the dielectric constant $\epsilon(\omega)$ and refractive index. The photocatalytic activity of TiO₂ is found to decrease with increasing pressure based on the absorption spectrum. The energy loss spectra show new peaks as a result of the pressure effect on the energy loss function. From our results, we noticed the effect of pressure within the range (0–60 GPa) on each of the structural, electronic and optical properties, and there is also good agreement between the current results and previous results.

Keywords: first principle; Ultrasoft pseudopotential; Density of states; Refractive index; Hydrostatic pressure.

ПЕРВИННЕ ДОСЛІДЖЕННЯ ЕФЕКТУ ВПЛИВУ ТИСКУ НА ЕЛЕКТРИЧНІ ТА ОПТИЧНІ ВЛАСТИВОСТІ РУТИЛУ TiO₂

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Анотація

Електронні константи, оптичні властивості та показники пружності рутилу TiO₂ досліджено з використанням першого принципу. Ефекти обмінної кореляції з використанням RPBE описано в термінах узагальненого градієнтного наближення [GGA]. Виміряні об'єм та об'ємний модуль добре узгоджуються з попередніми експериментальними і теоретичними результатами. Розраховано ширину забороненої зони (2.098 eV) з урахуванням більшої інтенсивності станів і розширеними енергетичними зонами навколо рівнів Фермі. Діелектричну проникність визначено з урахуванням електронної зонної структури і, з огляду на її важливість, використано для обчислення решти оптичних властивостей, таких як функція втрат енергії, показник заломлення, поглинання і відбивання. Вплив гідростатичного тиску (0–60 ГПа) показано на змінах властивостей. Результати нашого дослідження добре узгоджуються з відомими сучасними експериментальними і теоретичними результатами.

Ключові слова: перший принцип; надм'який псевдопотенціал; густина станів; показник заломлення; гідростатичний тиск.

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Introduction

Titanium dioxide (TiO₂) is a complex transition metal oxide that is readily available and well described experimentally, with a wide range of applications [1]. TiO₂ has a wide range of industrial applications, including photoelectrodes [2], electronics [3], gas sensing, painting [4], and dye-sensitized solar cells [5]. Hence, the utilization of TiO₂ in photocatalytic processes under visible light is significantly constrained. In order to enhance the range of wavelengths that TiO₂ can respond to and its overall photo catalytic efficiency across the entire solar spectrum, extensive endeavors have been undertaken [6–8]. Although many theoretical calculations of the electronic properties of rutile TiO₂ have been reported in the literature [9–14], there are significant differences between these previously calculated values and the corresponding experimental ones. The theoretical direct band gaps range from 1.67 to 3.25 eV (for the local density approximation, LDA), 1.69 to 4.45 eV (for the generalized gradient approximation, GGA), and more than 3.4 to 13.05 eV (for the Hartree-Fock approximation, HF). We do not know of any previous reports of a calculated, fundamental, indirect band gap for rutile TiO₂. Here, we report on studies of the electronic and structural band gap of rutile titanium dioxide at various hydrostatic pressure levels using a first-principles plane-wave ultra soft pseudo-potential method. Results and a discussion are presented in the section that follows, along with the specifics of the computational approach. The conclusion of this work is presented in the last section.

Theoretical Method

The present calculations are based on the implementation of the plane wave pseudo potential of density functional theory [15; 16] using the computational program CASTEP [17], which is supported by density functional theory and molecular dynamics theory. RPBE was first proposed in 1998 with the goal of improving in the chemical absorption of energy of atoms and molecules [18]. Therefore, we adopted GGA with the RPBE as the exchange correlation function [19]. The electronic arrangement of O and Ti are 2s²2p⁴ and 3s²3p⁶3d²4s², respectively. The valence of electrons O and Ti are 4 and 6 respectively. Ultrasoft Pseudo potential (USP) has been used to define the plane wave base and to perform mathematical operations to reduce the cutoff energy with minimal contribution from the core region. The kinetic energy cutoff (400eV) was used to extend the smooth portion of the wave function in the plane wave. A 2×2×2 k-point grid is used to integrate the Brillion region for rutile TiO₂. The Broyden-Fletcher-Goldfarb-Shenno (BFGS) miniature method was used to improve the geometry [20]. The convergence of displacement was set to less than 0.001 Å, and the energy difference was set to less than 1 × 10⁻⁶ eV / atom.

Results and discussion

Electronic properties. The BFGS algorithm was used to perform the geometry optimization. For the quadrangular rutile TiO₂ compound at the experimental lattice constant a = b = 4.716818 Å and c = 2.96673 Å. These equilibrium lattice constants are 1 % higher as compared to experimental values [21]. The equilibrium lattice state parametries, pressure derivative and bulk elastic modulus are presented below using Birch-Murnaghan equation [22]:

$$p(V) = 1.5B_0 \left[\left(\frac{V_0}{V} \right)^{\frac{7}{3}} - \left(\frac{V_0}{V} \right)^{\frac{5}{3}} \right] \left\{ 1 + 0.75(B' - 4) \left[\left(\frac{V_0}{V} \right)^{\frac{2}{3}} - 1 \right] \right\}, \quad (1)$$

where B' , B_0 and V_0 are pressure derivative of bulk modulus, bulk modulus and equilibrium volume, respectively. The data in Table 1 are

in good agreement with the theoretical [23–27] and experimental results [28].

Table 1

Calculate, volume (V_0) and bulk modulus (B_0) of rutileTiO ₂		
	V_0 [Å ³]	B_0 [GPa]
Present work	65.7	221
Zhu Jun et al. [23]	-	258
Iuga et al. [24]	63.3	235
Basavaraj et al. [25]	-	229.1
Yao et al. [LDA,GGA] [26]	61.11;64.47	245.2, 211.3
Zhou et al. [27]	64.34	221
Al-Khatatbeh et al. [28]Exp.	62.4	235

As shown in Fig. 1(a), the energy band gap was 2.098 eV, which is 40% smaller than that of the experimental band gap Ref [29].

The band gap calculation relies on the GGA approximation and DFT, suggesting that the band

gap magnitude is underestimated due to the derivative of the exchange-correlation energy gap [30; 31]. The energy band gap diagram of rutile TiO₂ is illustrated in Figure 1(a).

The top of valence band is chosen as the Fermi

level ($E_f = 0$ eV). The valence band maximum and conduction band minimum are observed at G point, which makes this material a direct gap (G-G) material. Some of the important features of this gap are presented in Table 2.

Table 2

Direct and indirect band gaps (eV) and their valence corresponding to the conduction band transfer of rutile TiO₂ at zero pressure

Valence to conduction band transition	Band gap energies (eV)
G-G	2.098
Z-Z	2.97
R-R	2.14
X-X	2.74
M-M	2.14
A-A	2.68

Fig. 1(b) shows the relationship between hydrostatic pressure and energy band gap with increasing pressure, the variable gap energy of rutile TiO₂ is indicated by the black dot in Fig. 1(b).

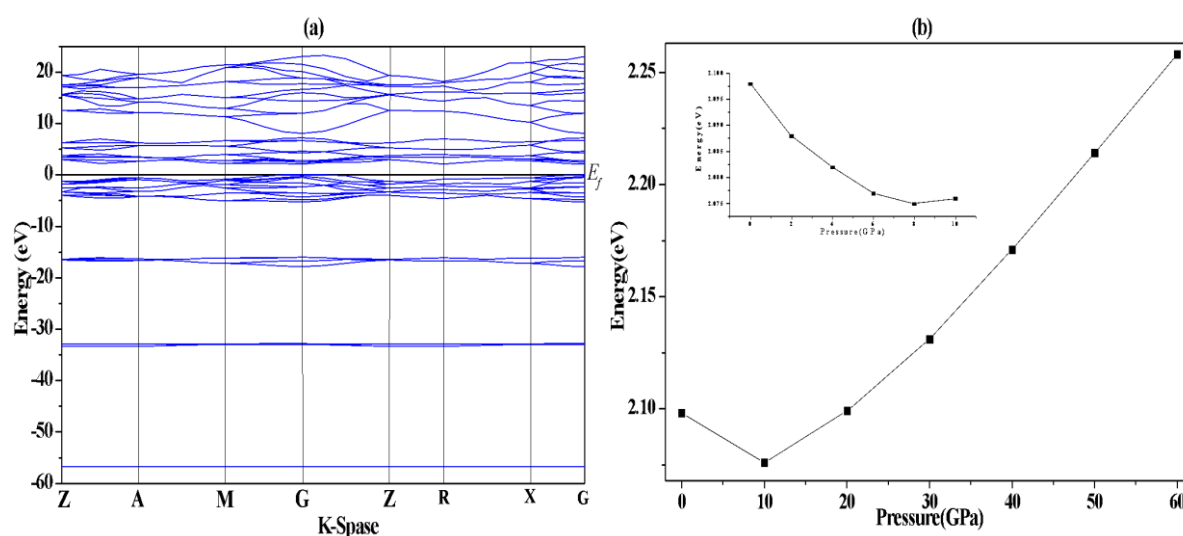


Fig. 1. a) Band structure of rutile TiO₂ at zero pressure and b) Relationship between band gap and pressure

This means that the energy gap passes through two stages: first stage falls within the range (0–8 GPa) of the applied pressure. The nonlinear

relationship between energy gap and pressure, which decreases with increasing pressure, can be fitted

using a third polynomial, which is expressed as:

$$E_g(p) = E_g(0) + 7.394 \times 10^{-4}p + 1.152 \times 10^{-4}p^2 + 1.563 \times 10^{-6}p^3. \quad (2)$$

In the second stage, after 10 GPa up to 60 GPa, energy gap increases almost linearly with increasing pressure in steps of 10 GPa. These differences may be related to slow changes in the crystal structure caused by the external pressure. The results showed that the energy gap related to different electronic states expanded with increasing pressure.

In order to get a better visualization of the electronic domain structure, T DOS and PDOS plots are shown in Fig. 2 (a–f). The total density of states roughly stretches from –57.5 eV to 23.3 eV, Fig. 2(a) determines the TDOS character with increasing pressure. Fig. 2(b–f) PDOS shows that the contribution of (O-2p)

states is dominant in VB, while (Ti-3d) states are less dominant for VB width it is 5.6 eV (fig. 2a), which are in agreement with the theoretical results of Baizae and Mousavi and the experimental results of 5.25 eV and 5.4 eV, respectively, [32; 33] from fig. 2 (b–f). As (Ti-d) and (O-p) states play an essential role in the construction of band conduction, from this we can conclude that transition between VB and CB is caused by the states of (O-2p) and (Ti-3d), since the states have maximum contribution to VB and CB. The DOS irregularity occurs in this region due to hybridization of the (O-p) orbital with the (Ti-3d) orbital. When DOS is compressed, it causes the energy gap to widen by shifting the valence band to

a lower energy level and the conduction band to a higher energy level. A decrease in TDOS

and PDOS indicates that fewer states are available for a specified level.

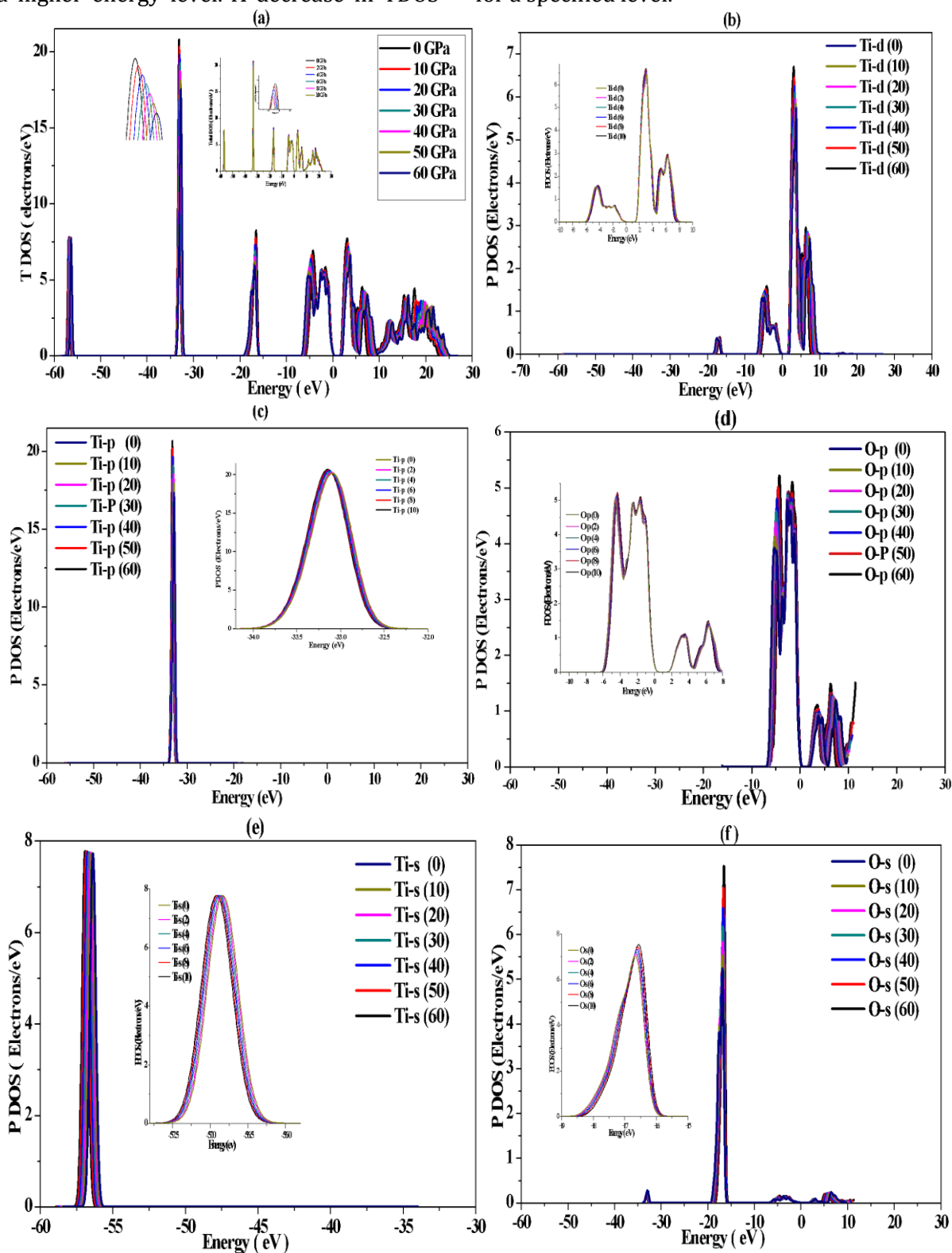


Fig. 2. a) TDOS of Rutile TiO_2 , and PDOS for b) Ti-d, c) Ti-p, d) O-p, e) Ti-s and f) O-s at zero and high hydrostatic pressures

Optical properties. The most important and widely studied are the optical properties of materials, which is closely related to electronic, magnetic and thermal properties. The optical properties including refractive index, dielectric

function, absorption coefficient and reflectivity obtained from light as it passes through the TiO_2 rutile were analyzed for the optimized TiO_2 rutile. Optical properties related to the complex dielectric function include conductivity, dielectric

constant, index of refraction, reflection and absorption. One is interconnected with the others. The electrical function is defined.

$$\varepsilon(\omega) = \varepsilon_1(\omega) + i\varepsilon_2(\omega), \quad (3)$$

where: $\varepsilon_1(\omega), \varepsilon_2(\omega)$ are express the real and

$$\varepsilon_2(h\omega) = \frac{2\pi e^2}{\Omega \varepsilon_0} \sum_{k,v,c} |\langle \psi_k^c | u, r | \psi_k^v \rangle|^2 \delta(E_k^c - E_k^v - E), \quad (4)$$

where ψ_k^c and ψ_k^v , μ, e represent the wave functions of the conduction band and valence band, incident electric field, and electronic charge at point k , respectively. The Kramer-kronig relation was used to obtain the real part from the imaginary dielectric function. From fig. 3(a), it can be seen that the imaginary part of the dielectric function it starts at 1.37 eV with a first peak of 5.12 eV due to excitation from VBM to CBM, second peak 7.69 eV and third peak 12 eV. This is due to transitions from A to M and M to G and G to L respectively. The influence of

imaginary parts of the dielectric function, respectively, in general, the dielectric function is associated with the electronic structure, the imaginary part of equation (2) is defined as follows:

hydrostatic pressure is manifested as a transformation of the imaginary part of the dielectric function (ε_2) towards larger values, as shown in Fig. 3(b). Light absorption efficiency of rutile decreases with increasing pressure. The dielectric function is important for the derivation of the other optical properties such as refractive index ($n(\omega)$), energy loss spectrum ($L(\omega)$), reflectivity ($R(\omega)$) and absorption coefficient ($\alpha(\omega)$) through the specified relationships [34–36].

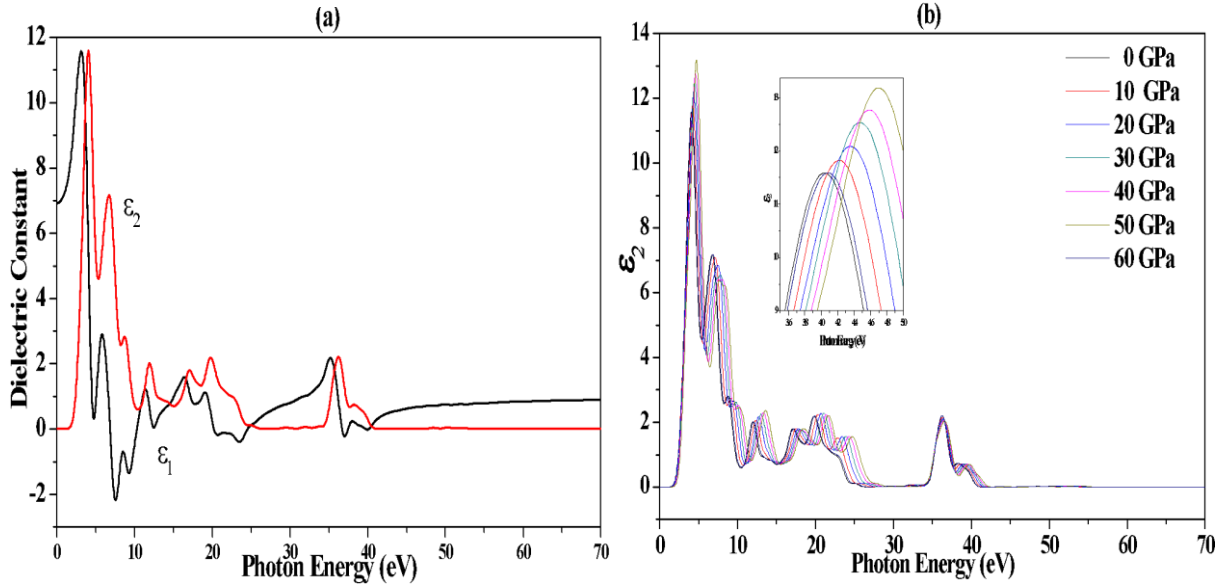


Fig. 3. Calculated a) real and imaginary part of dielectric constants at zero pressure, b) imaginary part of dielectric function (ε_2) under high pressure of rutile TiO_2

$$n(\omega) = \sqrt{\frac{|\varepsilon(\omega)| + \varepsilon_1(\omega)}{2}}; \quad (5)$$

$$L(\omega) = \text{Im} \left(\frac{-1}{\varepsilon(\omega)} \right) = \frac{\varepsilon_2(\omega)}{\varepsilon_1^2(\omega) + \varepsilon_2^2(\omega)}; \quad (6)$$

$$k(\omega) = \sqrt{\frac{|\varepsilon(\omega)| - \varepsilon_1(\omega)}{2}}; \quad (7)$$

$$R(\omega) = \frac{(n-1)^2 + k^2}{(n+1)^2 + k^2}; \quad (8)$$

$$\alpha(\omega) = \frac{2k\omega}{c}. \quad (9)$$

From fig.4(a) it can be seen that the refractive index is 2.63 and increases with pressure. Its peak is close to the value of 3.55 (under 60GPa) at an energy of 3.27eV in UV region as shown in Fig.4(a).

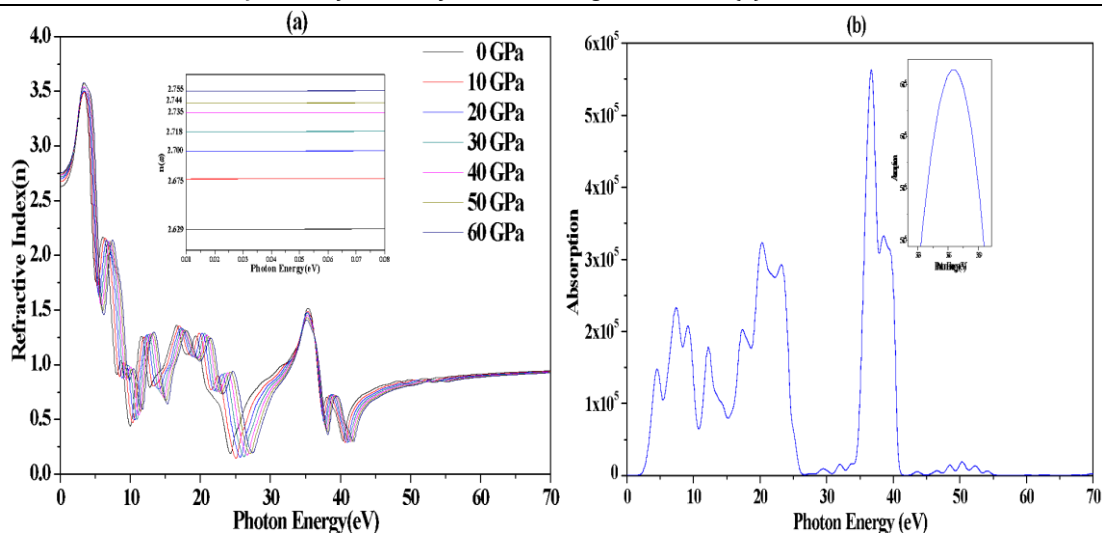


Fig. 4. Calculated a) refractive index($n(\omega)$) at various pressures b) the absorption coefficient ($\alpha(\omega)$) of rutile TiO_2

As for the results, they are summarized in Table 3 for the refractive index and dielectric constant. It has been compared with the existing literature [32; 37; 38]. It turns out that the value

of our calculated dielectric constant is somewhat greater compared to the values of the previous theoretical calculations [32] and experimental results [37].

Table 3

Calculated values of refractive index and dielectric constant of rutile TiO_2

Method	Refractive index (n)	Dielectric constant (ϵ_1)
GGA-RPBE [Present work]	2.63	6.91
GGA [32]	2.65	6.58
GGA+SOC [32]	2.65	-
LDA [32]	2.7	-
LDA+SOC [32]	2.71	-
Angel-Vosko [32]	2.54	-
GGA+PBE [28]	2.61	-
Experimental [37; 38]	2.71 [38]	6.693 [37]

While the value of the refractive index agrees with experimental results [38], Fig. 4(a,b) show the achieved results of refractive index($n(\omega)$), absorption coefficient($\alpha(\omega)$), and Fig. 5(a,b) show both the energy loss function($L(\omega)$) and reflectivity ($R(\omega)$) and, which are a major factor to ensure full energy utilization, and their main peaks are identified in the UV region as shown in Fig. 5(a). The energy loss function ($L(\omega)$) is a key factor to estimate the total energy utilization, its

main peaks are identified in the UV regions shown in Fig. 5(a). They are connected by rough edges in reflection spectra as shown in Fig. 5(b). From Fig. 5(b), it can be seen that rutile TiO_2 is an external absorbent of visible in accordance with the explanation of the study [39]. The absorption begins in the UV region (3.1 V) and reaches a maximum at 36.6 V, where the energy loss function is the lowest.

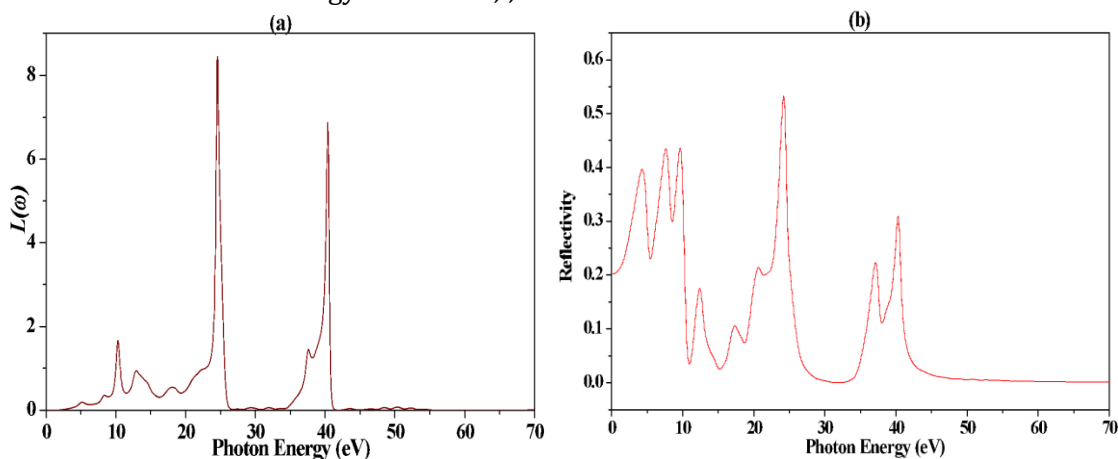


Fig. 5. Calculated a) the energy loss function ($L(\omega)$), b) the reflectivity ($R(\omega)$) of rutile TiO_2

Conclusion

We performed first principle calculations on rutile TiO₂ using the GGA-RPBE approximation combined with the Ultra soft pseudo potential method to investigate both electronic and optical properties. The approximate band gap value was 2.098 eV, which is smaller than the experimental value 3.0 eV, although this is in good agreement with other GGA-based calculations. On application of the external pressure, the direct band gap decreases in the first stage with the hydrostatic pressure from 0GPa to 8GPa and then increases in the second stage with the hydrostatic pressure from 10GPa to 60GPa. The valence bandwidth has been roughly determined to be 5.55 eV, which closely corresponds to the

experimental value of 5.4 eV and is consistent with other theoretical results as well. The dielectric constant was estimated to be 6.91, which is slightly over the experimental value 6.693. Whereas, the refractive index was determined with a value of 2.63 according to the electronic band structure against the experimental value of 2.71. The effects of pressure showed that the imaginary part of the dielectric function (ϵ_2) shifts towards higher photon energies as the complex refractive index decreases with increasing pressure. The absorption spectrum shows that TiO₂ poorly absorbs visible light and the absorption spectrum is dominated by ultraviolet light with a maximum at 36.4 eV.

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